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# **Cutting Indium Usage by 60% in SHJ-Modules Maintaining High Efficiency**

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**Abstract.** In this work the impact of reducing the Indium consumption for SHJ cells and modules is investigated. Optical simulations show that thinner Indium Tin Oxide (ITO) layers can be used on module level with minor reflection losses while on cell level losses are more severe. For extremely thin ITO layers with a thickness of 7-28 nm on texture a dielectric layer is necessary to maintain / improve the  $J_{SC}$  level on both cell and module levels. Results (i) on solar cells (Transfer Length Method - TLM) for lateral resistance and (ii) on shunt structures for vertical electrical resistance showed that there is significant improvement potential if the doping of the layers is adapted parallel to thickness reduction. Solar cell samples with 60% reduced ITO layer thickness on front and rear sides show a similar series resistance level as the 70 nm reference but lower *JSC*. Optical simulation showed that module integration will recover most of the lost  $J_{SC}$  resulting in an expected 0.5%<sub>abs.</sub> efficiency loss for samples with total (front & rear side) 60% less Indium without adding any process steps. Applying an additional dielectric film enables 80% Indium reduction on the front side with  $J_{SC}$  gain on cell level and similar  $J_{SC}$ on module level compared to the ITO with reference thickness.

**Keywords:** Silicon Heterojunction; Indium; Transparent Conductive Oxide

## **1. Introduction**

For silicon heterojunction solar cells (SHJ) the largest challenge is currently the consumption of scarce materials in the production that are critical for the technology to enter the TW market [1], [2], [3]. The following materials: Silver (Ag), Indium (In) and Bismuth (Bi) are the most relevant. The approach of this work to use less In in SHJ cells is to reduce the Indium-Tin-Oxide (ITO) thickness  $d_{\text{TO.txt}}$  [4], [5], [6], [7]. The ITO layer acts as transparent conductive oxide (TCO) on front and rear sides (FS/RS) of the SHJ cell. To compensate for increasing sheet resistance *Rsh* the effect of increasing the doping *Ne* of the thinner layer by varying the relative  $O_2$  flow during physical vapor deposition (PVD) is investigated. Besides the effect of  $d_{\text{TO,txt}}$  and *Ne* on the *Rsh* also the effect on the contact resistivities (i) between metal and ITO <sup>ρ</sup>*c,ITO,metal* as well as (ii) the FS stack consisting of the c-Si/a-Si:H(i)/a-Si:H(n)/ITO  $ρ<sub>c.c-Si.a-Si/TO</sub>$  [8], starting with the c-Si a-Si:H interface and finishing with the ITO bulk layer, is investigated. To understand about the differences in transparency of the TCO layers optical simulations were performed for the different cell and module scenarios with and without dielectric layer. The additional dielectric layer based on  $A<sub>2</sub>O<sub>3</sub>$  is applied to compensate for optical losses of thin ITO layers [4], [5].

# **2. Simulation of the short circuit current density potential**

To investigate the impact of FS ITO layer properties on the optics of a SHJ cell the online tool OPAL 2 from PV Lighthouse [9] was used to simulate four different scenarios for the nonmetallized cell stack: (i) with air as interface, (ii) with glass as interface, (iii) covered by a dielectric layer with air as interface and (iv) as (iii) with glass interface, as shown in Fig. 1. The simulated device stack consists of c-Si wafer as a base covered by a stack of 6 nm intrinsic and 6 nm n-doped a-Si:H layer and a ITO layer with varied doping  $(N_e \text{ from } 0.17 \times 10^{20} \text{ to } 10^{-10} \text{ m})$ 4.9x10<sup>20</sup> cm<sup>-3</sup>) and thickness  $(d_{\text{ITO.txt}}$  from 0 to 120 nm) on top. For the ITO simulation n and k parameters from [10] available in OPAL 2 [9] for a sputtered ITO film were used. The additional dielectric layer in scenario (iii) and (iv) is applied on top of ITO and metallization to reduce the optical losses due to increased reflection of samples with thin ITO layers. The dielectric layer thickness is optimized to maximize the sample's short circuit current potential: *JSC,pot* for each scenario and  $d_{ITO,txt}$ .



*Figure 1. The four cell / module scenarios.*

# **3. Experimental**

### **3.1 Symmetrically metallized solar cells and shunt structures**

To access the impact on the electrical parameters when changing the ITO properties rear emitter SHJ solar cells and shunt structures for the high-low FS contact were produced. Both sample structures are shown in the Fig. 2 below.



*Figure 2. Structures of SHJ cells and shunt structures.*

Both processes started with a cleaned and textured c-Si(n) wafer (base resistivity <sup>ρ</sup>*base* ~ 1 Ωcm) with *dW* ~ 165 µm thickness. The wafer surface was passivated by a stack of intrinsic and doped a-Si:H layers deposited by an OCTOPUS 2 PECVD tool from Indeotec. For the solar cells an a-Si:H(p) layer was deposited on the rear side (RS) to form the emitter while a- $Si:H(n)$  is deposited on the FS. In case of the shunt structures the a-Si: $H(n)$  was deposited on both sides symmetrically. After PECVD followed the ITO deposition with a Von Ardenne SCALA PVD tool equipped with a 97/3 rotary ITO target. Depositions were done at industry relevant speed at a power of 4.4 kW. For the SHJ cells on either FS or RS the reference ITO

layer was deposited while on the opposing side the ITO layer was varied in terms of thickness  $(d_{\text{TO.txt}}$  from 7 to 70 nm) and doping ( $N_e$  from 0.25 to 4.0x10<sup>20</sup>cm<sup>-3</sup>). For the shunt structures the varied ITO layers were deposited symmetrically on RS and FS. Subsequent as a last step followed metallization of the samples. The solar cells were symmetrically metallized on an ASYS Ekra screen printing line applying a fine line screen with an opening  $w_N$  of 20  $\mu$ m, resulting in printed width  $w_F$  of 30.6  $\mu$ m  $\pm$  2.2  $\mu$ m and a pitch  $p_F$  of 1.3 mm while the shunt structures were metallized with a narrower  $p_F = 0.65$  mm and wider opening  $w_N = 40$  µm ( $w_F$ about 51 µm) to keep the resistive losses due to lateral transport in the ITO and at the ITOmetal contact low. In Fig 3. the process of the cell production is documented.

| Group                    |   |             | 2        | 3 | 4              | 5 | 6         | 7   | 8             | 9 | 10             | 11 | 12       |
|--------------------------|---|-------------|----------|---|----------------|---|-----------|-----|---------------|---|----------------|----|----------|
| side                     | Rear emitter SHJ prec. n/i/N/i/p after Texture, Cleaning and PECVD  |             |          |   |                |   |           |     |               |   |                |    |          |
|                          | $O2$ flow var.  | Reference   |          |   |                |   |           | Low |               |   |                |    | High     |
| <b>RS</b><br><b>Mask</b> | $d_{\text{ITO.txt}}$ [nm]   |             |          |   |                |   |           |     | 7<br>13<br>28 |   | 13<br>28<br>70 |    | 13<br>28 |
|                          |   | 70          |          |   |                |   |           |     |               |   |                |    |          |
|                          | $O2$ flow var.  | High<br>Low |          |   |                |   | Reference |     |               |   |                |    |          |
| FS.                      | $\overline{\mathsf{d}_{\mathsf{ITO},\mathsf{txt}}}$ [nm]  |             | 13<br>28 |   | 13<br>28<br>70 |   | 13<br>28  |     | 70            |   |                |    |          |
| metal                    | Symmetrical LTP metallization (1.3 mm pitch 0BB)   Drying 200°C 1min   Curing 220°C 1min                                    |             |          |   |                |   |           |     |               |   |                |    |          |
| char                     | Hall measurement ( $\mu$ , N <sub>e</sub> ) Cell tester: FS + RS IV GT®; TLM: R <sub>sh</sub> and $\rho_c$ for RS variation |             |          |   |                |   |           |     |               |   |                |    |          |

*Figure 3. Process flow for solar cell production.*

Both cells and shunt structures were then characterized in a HALM IV cell tester equipped with a GridTouch® contacting unit. For cells the complete set of IV parameters was extracted while for the shunt structures only the shunt resistance  $R_P$  was extracted. The  $R_P$  is the base to determine the contact resistivity of the layer stack c-Si/a-Si/ITO <sup>ρ</sup>*c,c-Si,a-Si,ITO* according to equation (1) by subtracting the wafer bulk contribution, dividing by a factor of 2 and finally correcting for the ITO-metal contact <sup>ρ</sup>*c,ITO,metal* that is used while taking into account the metallized area fraction *AM*. Whereas the corrections for wafer are significant, the metal ITO contact correction shows only a minor impact due to the high *AM* applied for the shunt structures.

$$
\rho_{c,c-Si,a-Si,ITO} = \frac{R_P - \rho_{base} \cdot d_w}{2} - A_M \cdot \rho_{c,ITO,metal} \quad (1)
$$

To access the <sup>ρ</sup>*c,ITO,metal* in the equation above and the *Rsh* of the different ITO layers TLM measurements were done on the emitter side of SHJ cells with varied RS ITO. To verify the results from the optical simulations, solar cell samples from the reference group 4 with varied  $d_{\text{ITO}}$ <sub>txt</sub> for a fixed  $O_2$  flow received a thermal ALD Al<sub>2</sub>O<sub>3</sub> layer on top of ITO and metal and were measured again after the deposition. All ITO layers with different thicknesses and doping were also deposited on flat glass substrates and then characterized with a Hall setup. The parameters doping *Ne* and mobility *µ* were extracted.

# **4. Results**



*Figure 4. Photo of the thickness variation applied during ITO deposition.*

Fig. 4 above visualizes the different applied  $d_{TOL,kt}$  that are shown for the masked RS ITO of SHJ cells, also the approximate relative  $d_{\text{ITO.txt}}$  reduction between 60% and 90% compared to the reference with  $d_{\text{ITO.txt}}$  = 70 nm are mentioned.

## **4.1 Short circuit current density potential simulation**

In Fig. 5 the results of the optical simulation for the  $J_{SC, pot}$  that is proportional to the short circuit current density *JSC* of cells or modules for the four scenarios (Fig. 1) is shown. Reducing the  $d_{\text{ITO,txt}}$  on the FS of solar cells measured in air (turquoise data points in Fig 5 left) leads to a fast decline of the *JSC,pot* of the SHJ cell due to the rapidly increasing reflection when moving away from the optimum around  $d_{\text{TO,txt}}$  = 70 mm. At the optimum reflection is low because the film has antireflective (AR) properties. For cells encapsulated in modules (blue data) the impact of reduced  $d_{\text{ITO,txt}}$  is less pronounced (max. J<sub>SC</sub> loss ~ 1 mA/cm<sup>2</sup> in the module versus ~ 3 mA/cm<sup>2</sup> in air). The graph right in Fig. 5 shows the impact of  $d_{\text{ITO.txt}}$  with an additional dielectric layer on the cell (orange) and module level (red). For moderate doping the  $J_{SC, pot}$  is rather independent of  $d_{\text{TO.txt}}$  since the dielectric layer thickness is optimized for maximum  $J_{\text{SC,pot}}$  and then acts as primary AR layer. For moderate *N<sub>e</sub>* highest *J<sub>SC,pot</sub>* are found with an optimal  $d_{ITO,txt}$  of 30-40 nm for cells and 40-50 nm for modules. For high  $N_e$  of 4.9x10<sup>20</sup> cm<sup>-3</sup>  $J_{\rm SC}$  degrades quickly with increasing  $d_{\text{ITO.txt}}$  due to increased free carrier absorption (FCA) in the ITO layer.



*Figure 5. Simulated short circuit current potential of FS ITO layers with varied thicknesses and doping for the four scenarios: versus air and glass interface (i) for ITO films (left) and (ii) for ITO capped by a dielectric layer (right).*

### **4.2 Effect of dielectric layer on short circuit current density of SHJ cells**

To verify the simulated  $J_{SC,pot}$  increase when a dielectric layer is added ontop of ITO layers with varied  $d_{\text{TO,txt}}$ , samples with varied FS ITO from group 4 where further processed. First, they were measured initially then a dielectric  $Al_2O_3$  layer was deposited on top of the ITO. The dielectric layer thickness was optimized for each  $d_{\text{ITO.txt}}$  individually for highest  $J_{\text{SC,pot}}$  in air. Finally, the samples were measured a second time. The difference in  $J_{SC}$  was calculated and compared to the *JSC,pot* increase from the simulations. The result is shown in Fig. 6. The simulation and experimental data show close to perfect agreement. Further IV data is not presented here since the samples with capping were difficult to contact with the busbarless contacting unit and a reduced FF was measured. Similar samples were successfully integrated into modules [11] showing that capping with a dielectric layer is no fundamental problem.



*Figure 6. Comparison of simulated with measured optical gains when an optimized dielectric capping is deposited on an ITO with varied thicknesses.*



#### **4.3 TLM measurements**

*Figure 7. Results from TLM measurement: (a) sheet resistance Rsh, (b) sheet bulk resistivity* <sup>ρ</sup>*bulk,ITO and (c) contact resistivity ITO-metal* ρ*c,c-Si,a-Si,ITO.* 

The Fig. 7 above shows the extracted  $R_{sh}$  from TLM measurements for the different  $N_e$ levels and  $d_{\text{ITO,txt}}$  ranging from 7 nm to 70 nm where corresponding ITO saving range from 0% for the reference to 60, 80 or 90%. For ≥ 80% reduced  $d_{ITO.txt}$  the  $R_{sh}$  increases quickly. Comparing to the reference layer (blue) shows that for 60%  $d_{\text{ITO,txt}}$  reduction only a minor increase in *Rsh* is found. Looking at the calculated bulk ITO resistivity <sup>ρ</sup>*bulk,ITO* (Fig. 7 b) reveals that for 60% or 80% reduced *dITO,txt* and in parallel increased *Ne* the <sup>ρ</sup>*bulk,ITO* is improved by a factor larger than 2. This means that the In utilization can be significantly improved compared to the reference. Contact resistivity <sup>ρ</sup>*c,ITO,metal* data, here shown only indicative from linear fits (arbitrary units) due to inconsistency of the TLM-analysed data gathered with different software versions, show similar trends as  $R_{sh}$  and  $\rho_{bulk,ITO}$ . Reducing the  $d_{ITO,txt}$  leads to an increase of the resistance but this can be (partially) compensated by increasing *Ne*. The TLM analysis shows: in terms of contact to the metal and lateral resistance reducing the  $d_{\text{ITO,txt}}$  and increasing  $N_e$  in parallel leads to improved utilization of In while *Rsh* and <sup>ρ</sup>*c,ITO,metal* data are only increased marginally for 60% but more pronounced for 80% or 90% thickness reduction.

### **4.4 Cell results**

A selection of the cell IV parameters is presented in the following figures. The  $J_{SC}$  in Fig. 8 shows the effect of increasing reflection of cells measured with air as interface when  $d_{\text{TO.txt}}$  on the FS is reduced. The results for the lowest  $O_2$  flow show deviant behaviour due to changed morphology of the ITO-film and is excluded from the discussion. Compared to the simulation (Fig. 5) overall J<sub>SC</sub> reduction ( $d_{\text{TO,txt}}$  70 nm  $\rightarrow$  7 nm) is with ~ 3mA/cm<sup>2</sup> similar. For  $d_{\text{TO,txt}}$  = 28 nm an impact of the *Ne* variation is visible, for 14 nm an insignificant trend is found while for 7 nm a trend is not visible anymore. For the RS only a minor impact of  $d_{ITO,txt}$  and  $N_e$  was found. In tendency thinner layers on the RS lead to increased  $J_{SC}$  measured on the FS. Further also higher  $N_e$  led to increased  $J_{\text{SC}}$ .



**Figure 8.** Cell tester results for the  $J_{SC}$  with varied  $d_{IO,txt}$  on cell FS and RS.

In Fig. 9 the  $R_s$  is reduced for a given  $d_{\text{ITO,txt}}$  when  $N_e$  is increased for both FS and RS deposition (besides the excluded group with lowest  $O<sub>2</sub>$  flow). Compared to the reference 70 nm ITO on the FS (Fig. 9 left) a 28 nm  $d_{TO,txt}$  shows an overall reduced  $R_S$  when  $N_e$  is adapted. Even though  $R_{sh}$  was increased for 14 nm  $d_{\text{TO.txt}}$  the reference  $R_s$  level can be maintained. For 7 nm *d<sub>ITO,txt</sub>* the *R<sub>S</sub>* level is increased by 0.3 Ωcm<sup>2</sup> for high *N<sub>e</sub>*. On the RS (Fig. 9 right) the findings are similar to the FS, but the  $R<sub>S</sub>$  level increases more quickly with reduced  $d_{(TO, tot)}$  due to the stronger impact of the increasing *Rsh* on the emitter side. For 28 nm the reference *RS* level can be kept constant, while for 14 nm or 7 nm it increases by 0.2 or 0.6 Ωcm<sup>2</sup> respectively when  $N_e$  is optimized. The  $d_{\text{ITO,txt}}$  on the RS can be reduced by 60% without increasing resistive losses. In the considered cases the effect of increased *Rsh* on the RS is pronounced due to the large *p<sub>F</sub>* of 1.3 mm that is typically chosen smaller on the RS to improve η. But overall decreasing the  $p_F$  is an option to compensate for higher  $R_{sh}$  related losses both on FS and RS.



*Figure 9. Cell tester results for the series resistance RS.*

#### **4.5 Impact of lateral and vertical resistance on the series resistance**

In the following Fig. 10, the  $R_{sh}$  results from TLM and the  $\rho_{c,c-Si,a-Si/TO}$  determined from shunt structures are plotted versus the *RS* of the symmetrically metallized solar cells with single sided ITO variation.



*Figure 10. Scatter plots of R<sub>S</sub> versus R<sub>sh</sub> for ITO variation on cell FS and RS (left) and versus*  $\rho_{c.c-Si.a.}$ *Si,ITO for the cell FS (right).*

As shown in section 3.1 the  $R_{sh}$  increases when  $d_{170, tot}$  is reduced while adapting/increasing *Ne* leads to decreased *Rsh*. The effect on *RS* shown in the left of Fig. 10 is more pronounced on the emitter side of the cells. Low *RS* levels are achieved for 28 nm on FS and RS and 14 nm *d<sub>ITO,txt</sub>* on the FS. Best cells with 7 nm on the FS and a  $R_{sh} \sim 1000$  Ω/sq led to an  $R_s$ increase of 0.2 Ωcm² while on the RS the *RS* increase is with 0.6 Ωcm² much more pronounced for the same  $N_e$ . The impact of the increased  $R_{sh}$  can be compensated by reducing the  $p_F$ . Here *RS* and shading losses need to be balanced on the FS for optimal η while on the RS shading effects are less relevant. For the FS <sup>ρ</sup>*c,c-Si,a-Si,ITO* investigation many samples were broken during the process and only a few samples were finally measured at the cell tester (Fig. 10 right). Compared to the reference a 28 nm ITO leads to improved <sup>ρ</sup>*c,c-Si,a-Si,ITO*, adapting the *N<sub>e</sub>* had only a small impact on the  $ρ<sub>c.c-Si.a-Si,ITO</sub>$  but helped for the 14 and 7 nm layers on the FS to keep <sup>ρ</sup>*c,c-Si,a-Si,ITO* low. The <sup>ρ</sup>*c,c-Si,a-Si,ITO* needs to be reviewed critically in the future since current crowding effects for high *Rsh* values cannot be excluded and potentially led to a substantial effect of the  $p_F$  when the current is forced to flow in the wafer and only directly under the contact finger from the wafer through the layer stack into the metal finger. The data show that even for 7 nm thick layers with  $ρ<sub>c,c-Si,a-Si,ITO</sub> ~ 0.23$  Ωcm<sup>2</sup> acceptable  $R<sub>S</sub>$  increase of 0.2 Ωcm<sup>2</sup> were achieved on the FS when *Ne* is increased while without adapting the *Ne* the *RS* increases by  $0.45$  Ωcm<sup>2</sup>.

#### **4.6 Expected efficiency level**



*Figure 11. Expected change in cell and module efficiency for samples with dielectric capping compared to the uncapped reference with variation of the ITO thicknesses on the FS.* 

Based on the IV results of the symmetrically metallized solar cells and the simulated change in *JSC,pot* an expected *η*-level, see Fig. 11, was roughly calculated. Here the *η* on cell level for the reference ITO and varied  $d_{\text{ITO,txt}}$  on the FS with adapted  $N_e$  was considered as well as the expected change in  $J_{SC}$  according to OPAL 2 simulation (compare Fig. 5 and 6). On cell level reducing  $d_{\text{TO.txt}}$  on the FS by 60% or more leads to high *η*-loss (>1%<sub>abs</sub>). On module level for the 60%  $d_{\text{TO,txt}}$  reduction the expected *η*-loss is 0.5 %<sub>abs</sub>. If an additional dielectric layer is applied on a 28 nm ITO, *η* is expected to improve on cell level (~0.3%<sub>abs</sub>) compared to the reference ITO without dielectric layer. On module level with a capping layer the *η* can be maintained or marginally improved. Reducing  $d_{ITO,txt}$  by 90% leads to significant losses in all scenarios due to decreasing  $J_{SC}$  and high  $R_S$ . This investigation does not consider the adaption of the metallization to compensate for an increase in lateral  $R_{sh}$  losses by applying a smaller  $p_F$ . Here it is expected that the *η*-optimum shifts to thinner ITO layers as long as the *JSC,pot* (Fig. 5) is not substantially declining. On the cell RS applying a 28 nm  $d_{(TO,txt)}$  was possible while keeping the same  $R<sub>S</sub>$  level. In summary  $d_{\text{1TO}, \text{txt}}$  reduction on both sides by 60% is possible on module level when 0.5 %<sub>abs</sub> *η*-loss is considered acceptable. If an additional dielectric layer is applied the FS *dITO,txt* reduction can be increased to 80% without scarifying *η* on both cell and module level.

## **5. Conclusion and Outlook**

Within this work promising ITO layers with reduced  $d_{\text{ITO,txt}}$  were explored that allow to reduce In-consumption by 60% or more. On the cell RS a 60% reduction was shown to achieve the same  $R_s$  level as the 70 nm reference even though the  $R_{sh}$  increased, and a large  $p_F$  was applied. Since the optics on the RS are less important [10], both for bifacial but especially for mono-facial modules thickness reduction by 60% or more are expected to be applicable without *η* loss. On the cell FS the *RS* level can be improved when going from 70 nm reference to 28 nm but higher reflection leads to *η*-loss greater than 1%abs on cell and an expected 0.5%abs loss on module level. Adding a dielectric layer on the FS enables *η*-gains on cell level and similar η-level in modules for 28 nm thick ITO films compared to the reference. Also capped thin ITO layers (14 nm / 80%) are expected to achieve similar *η* on cell and module

level. Central factors effecting the R<sub>S</sub> were explored by TLM and shunt structure measurements. The interplay of both extracted parameters *Rsh* and <sup>ρ</sup>*c,c-Si,a-Si,ITO* were shown to be useful to describe and understand  $R<sub>S</sub>$  related effects. It could be shown that the In utilization in terms of  $\rho_{bulk,ITO}$  can be substantial improved if  $d_{ITO,txt}$  and  $N_e$  are both adapted. The process sequence for shunt structures needs more attention in the future. It is planned to investigate current crowding and spreading resistance effects to clarify if the method with a printed grid is reliable also for ITO films with high *Rsh*. For thin ITO layers and additional dielectric layers module integration and module reliability needs to be investigated. Besides the approach of capping thin ITO with a dielectric layer also the simplified approach with a thin In-based TCO capped with In-free TCO is investigated. This allows strong reduction of the In-consumption and simplifies the processing since no extra tool for applying the capping layer is necessary [12], [13].

# **Data availability statement**

The data supporting the results of this contribution are available upon reasonable request from the corresponding author.

# **Author contributions**

Conceptualization: S. Pingel, M. Bivour, A. Steinmetz; Funding acquisition: M. Bivour, A. Steinmetz; Methodology: S. Pingel, A. Krieg, M. Bivour; Project administration: S. Pingel, I. Vulcanean, W. Wolke; Resources: I. Vulcanean, C. Röhnelt, W. Wolke, V. Georgiou-Sarlikiotis; Supervision: A. Steinmetz; Writing – original draft / review & editing: S. Pingel, A. Krieg, M. Bivour, A. Steinmetz

# **Competing interests**

The authors declare that they have no competing interests.

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